

## Research Article

# **Modulation Electronic Properties of Silicane/SnSe<sub>2</sub> Van der Waals Heterostructures Using External Force and Electric Field**

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In recent years, much interest in the study of Van der Waals heterostructures (vdWhs) has arisen. This has led to a significant amount of fundamental research being produced, from which novel optoelectronic applications have been established. By using first principles, we analyze the electronic structure of silicane/SnSe<sub>2</sub> vdWhs in the response to an externally applied electric field and a normal strain. The results show that the silicane/SnSe<sub>2</sub> vdWh acts as an indirect semiconductor when it is subjected to an applied electric field between -1 and 0.1 V/Å and becomes a metal in the 0.2 to 1 V/Å range. Significantly, the electronic band alignments of the silicane/SnSe<sub>2</sub> vdWhs are modified from a type-II to a type-I when a field of -0.7 V/Å is applied. Furthermore, it is determined that the silicane/SnSe<sub>2</sub> vdWhs appears to have a semiconductor-metal phase transition at a strain of -5%. Our results indicate that the silicane/SnSe<sub>2</sub> vdWhs have the potential for applications in novel high-performance optoelectronic devices.

### 1. Introduction

Graphene, a flat mesh of hexagonal carbon atoms, was first successfully synthesized in 2004 by Novoselov et al. [1]. This was a significant step forward in the research of materials and assisted the movement of the field from the realm of the three-dimensional (3D) scale to two dimensions (2D). Since the creation of graphene, a growing number of unique 2D materials have been predicted theoretically and later synthesized experimentally, such as silicene [2-5], silicane [6–9], phosphorene [10, 11], hexagonal boron nitride [12], transition-metal dichalcogenides [13-17] (TMDs), and monolayer SnSe<sub>2</sub> [18-22]. However, no single 2D material possesses all the useful properties that are required for functional application in all areas. For example, silicene has limited capabilities within switching devices because of its zero-bandgap characteristics; in contrast, a typical insulator boron nitride is useful in such an application due to its very large bandgap [23-25]. Therefore, depending on the sought application, the electronic structures of existing two-dimensional materials need to be modified for each scenario.

Fortunately, it has been found that the vertical stacking of two different two-dimensional materials can enable the formation of Van der Waals heterostructures (vdWhs), which not only overcomes the drawbacks of the individual materials, but also generates unusual properties and new phenomena due to the coupling between the layers [26–29]. This discover has extended the possible applications of twodimensional materials for novel next-generation optoelectronic devices.

Silicane, i.e., fully hydrogenated silicene [30], has a large bandgap and a high carrier mobility [8, 9], and it is a substance that has been extensively studied since it was first synthesized by Yamanaka et al. [31]. Restrepo et al. [32] found that the application of a strain can modify the electronic structure and increase the electron mobility in silicane. Gang et al. [33] determined that the electronic structure of silicane is tunable via the application of a uniform tensile strain and an electric field. Wang and Yu [34] discovered that the bandgap of silicane can be modulated continuously by use of an applied electric field and a biaxial tensile strain. Thus, it is clear from these studies that an applied electronic field and a strain can adjust the electronic structures of silicane. Monolayer SnSe<sub>2</sub>, a typical group IV-VI semiconductor that contains a bandgap of 1.38 eV, is widely used in applications such as high-speed photodetection, phase-change memory, and field-effect transistors. Gonzalez and Oleynik [21] determined that the monolayer SnSe<sub>2</sub> possesses an indirect bandgap. Huang et al. [35] showed that the monolayer SnSe<sub>2</sub> sheet has a semiconductor-metal transition when placed under strain. Hien et al. [36] found that with an increasing tensile biaxial strain the bandgap of monolayer SnSe<sub>2</sub> raises slightly. Xia et al. [37] discovered that phosphorene/SnSe2 vdW heterostructures possess a type-III band alignment and provides a negative differential resistance effect. These investigations demonstrated that the electronic structures of silicane and SnSe<sub>2</sub> can be modified by application of an electric field and a strain and by the creation of a vdWhs. However, there has been insufficient research in this field, particularly for silicane/SnSe2 vdW heterostructures. In this work, by applying both an electric field and a biaxial tensile strain, we analyze the electronic structure of silicane/SnSe<sub>2</sub> vdW heterostructures via density functional theory (DFT). We verify that the applied field and the strain can modify the electronic structure of the silicane/SnSe<sub>2</sub> vdW heterostructure.

#### 2. Computational Methods

The computational methods of this paper are performed using calculations based on first principles and DFT. For the crystal structure relaxation and the electronic structure calculations, the generalized gradient approximation (GGA) of the Perdew-Burke-Ernzerhof (PBE) functional [38, 39] is utilized, which is part of the SIESTA code [40, 41]. The PseudoDojo method is employed to solve the Kohn-Sham equations. In these calculations, the energy cut-off is chosen as 100 Hartree. A Brillouin zone, which is useful in crystal structure relaxation, is sampled in the K-mesh of  $9 \times 9 \times 1$ , while a K-mesh of  $11 \times 11 \times 1$  is utilized for calculations on the energy band structure. For the relaxation, the convergence energy criteria are chosen as  $10^{-4}$  eV, and the residual force is lower than -0.01 eV/Å. To avoid unnecessary interactions between neighboring structures, the thickness of the vacuum layer is chosen to be 20 Å. Significantly, holding the mutual positions of the two material layers together requires a long-ranged vdW interaction. Here, to account for these long-range electron correlation effects, the vdW-DF3 functional is the one chosen [42].

#### 3. Results and Discussion

To begin, the geometries of the pristine silicane and the  $SnSe_2$  monolayers are optimized. For the pristine silicane monolayer, the relaxed lattice constants are determined as a = b = 3.889 Å; these value are consistent with the values determined from theory that are found in previous work [30, 33, 34]. The optimized lattice constants for the pristine  $SnSe_2$  monolayer are a = b = 3.921 Å, and again this agrees with the literature values [21, 35, 36]. These results are an indication that our methods are both acceptable and

produce reliable calculations. Next, we investigate the electronic properties of pristine silicane and the SnSe<sub>2</sub> monolayers. The pristine silicane monolayer band structure, which is calculated by use of the PBE functional, is shown in Figure 1(a). Here, it is observed that the conduction band minimum (CBM) and the valence band maximum (VBM) are positioned at the M point and the  $\Gamma$  point, respectively. This signifies that pristine silicane is an indirect bandgap semiconductor, which contains a bandgap of 2.184 eV; this is consistent with the studies of Wang and Guzmán-Verri [34, 43]. Figure 1(b) shows the band structure that is calculated using the PBE functional of pristine SnSe2. The CBM and the VBM are found at the M point and the  $\Gamma \longrightarrow M$  path in the two-dimensional hexagonal Brillouin zone, respectively. This indicates that pristine SnSe<sub>2</sub> is also an indirect bandgap semiconductor, which has a bandgap of 0.963 eV that is similar to the value (0.85 eV) determined by Li et al. [44]. The small error can be attributed to the different software packages used for the calculations. In addition, we also observe that our results are smaller than the analogous values (2.94 eV for silicane [34]; 1.46 eV for SnSe<sub>2</sub> [45]) that are obtained using the HSE06 method; an explanation is the possible underestimation of the bandgap by the PBE method. The correct band structure trend, and a proper understanding of the physical mechanisms, can be predicted by the standard PBE functional. To save on computational expenses, we thus use such a procedure to execute the electronic structure calculations for the silicane/SnS<sub>2</sub> vdW heterostructure for both an applied electric field and a strain.

To attain the minimum lattice mismatch value, a unit cell of  $SnSe_2$  and monolayer silicane are used to construct the silicane/ $SnSe_2$ ; the determined lattice mismatch is only ~0.8%. Compared with previous studies [22, 46–48], the obtained lattice mismatch value is extremely small, which indicates that the silicane/ $SnSe_2$  vdW heterostructure can be constructed with ease. To determine the most stable structure, we consider six possible heterostructure stackings that are labeled as Model-I to Model-VI and correspondingly shown in Figures 2(a)–2(f).

To determine the vdW interactions between the silicane and  $SnSe_2$  monolayers, the interface distance between the two layers (symbolized by  $\Delta$ ) and the formation energy ( $E_f$ ) of the silicane/SnSe<sub>2</sub> vdW heterostructure are determined from the following equation:

$$E_f = E_{\text{total}} - E_{\text{sili}} - E_{\text{SnSe2}},\tag{1}$$

where  $E_{\text{total}}$ ,  $E_{\text{sili}}$ , and  $E_{\text{SnSe2}}$  are the total energy of the silicane/SnSe<sub>2</sub> vdW heterostructure, silicane, and SnSe<sub>2</sub> monolayer, respectively. The values calculated for  $E_{\text{f}}$  and  $\Delta$  are provided in Table 1. Since all the calculated values for  $E_{\text{f}}$  are negative, we can confirm that the six stacking configurations are energetically stable and the vdW interactions between the silicane and SnSe<sub>2</sub> layers do exist. On examining all the configurations, we determine that the conformation Model-I is the most stable structure since it has the lowest formation energy and the smallest interfacial distance. For this reason, we solely focus on this model in the following text.



FIGURE 1: Graphs showing the band structures (in terms of units of energy) of (a) the silicane and the (b)  $SnSe_2$  monolayers. (c) The projected band structures of the silicane/ $SnSe_2$  vdW heterostructure are also given. Here, the dull-red lines represent  $SnSe_2$ , while the dark blue lines denote silicane. (d) A depiction of the band offset of the silicane/ $SnSe_2$  vdW heterostructure is also provided.

Now, based on the PBE functional method, we calculate the electronic structures of the silicane/SnSe<sub>2</sub> vdW heterostructure (Model-I); the results are shown in Figure 1(c). The VBM predominantly derives from the silicane monolayer, and it is found at the  $\Gamma$  point, while the CBM principally originates from the SnSe<sub>2</sub> monolayer and it is positioned at the M point. From these outcomes, we can establish that the material is an indirect semiconductor that contains a bandgap value of 0.133 eV and has a type-II band alignment [49]. The bandgap of the silicane/SnSe<sub>2</sub> vdW heterostructure is less than those of the pristine silicane monolayer and the SnS<sub>2</sub> monolayer, which indicates that the formation of the silicane/SnSe<sub>2</sub> vdW heterostructure can effectively modify the bandgap. To analyze the electronic structures of the silicane/SnSe2 vdW heterostructure in greater depth, we now draw the energy band alignments of the silicane/SnSe2 vdW heterostructures, as shown in Figure 1(d). The band edges of silicane are (2.151, -0.041) eV, while the corresponding values for SnSe<sub>2</sub> are (0.092, -0.858) eV. The CBM and VBM of the silicane layer are both higher than those of the SnSe<sub>2</sub> layer, which further confirms that the silicane/SnSe2 vdW heterostructure has type-II characteristics [49]. These discoveries mean, due to the typeII band alignment and the suitable bandgap value, that the silicane/ $SnSe_2$  vdW heterostructure can be employed in high-efficiency infrared sensors.

Previous studies show that on application of an external electric field onto a vdW heterostructure system the electronic structure of the material can be substantially altered [50, 51]. Hence, under the influence of a vertical electric field, we now analyze the electronic structure of the silicane/SnSe<sub>2</sub> vdW heterostructure. The path of the electric field is chosen to be perpendicular to the length of the silicane/SnSe<sub>2</sub> vdW heterostructure: the positive direction is presumed to be from the silicane towards SnSe2. The electric field is implemented within the range of -1.0 to 1 V/Å in steps of 0.1 V/Å. The effects of the electric field on the band edges, the bandgap, and the band structures of the silicane/SnSe2 vdW heterostructure are illustrated in Figure 3 and Figure 1S. We determine that, in the range of -1 to 0.1 V/Å, the silicane/ SnSe2 vdW heterostructure is an indirect semiconductor, while it is a metal for 0.1 to 1 V/Å. In addition, for -1 to -0.7 V/A, the CBM and VBM are both seen to mostly originate from the monolayer silicane, which signifies a type-I band alignment. In the range between -0.7 and 0.1 V/Å, the CBM mostly originates from the monolayer silicane, while



FIGURE 2: Depictions of the lattice structures  $(2\times 2)$  of the silicane/SnSe<sub>2</sub> vdW heterostructure, in which (a) Model-I, (b) Model-II, (c) Model-II, (d) Model-IV, (e) Model-V, and (f) Model-VI are illustrated. The upper diagrams show the top view, while the lower ones are the side view. The unit cells are represented by the black lines.

TABLE 1: The formation energy  $(E_f)$  of the heterostructure and the interface distance between the two layers ( $\Delta$ ) of the six stacking conformations.

Model	Ι	II	III	IV	V	VI
$E_f$ (eV)	-0.720	-0.692	-0.688	-0.692	-0.679	-0.690
Δ (Å)	2.938	3.185	2.969	3.245	3.242	3.235



FIGURE 3: Graphs showing (a) the band edge and (b) the bandgap as a function of the external electric field strength that is applied to the silicane/SnSe<sub>2</sub> vdW heterostructure.

the VBM is primarily from the SnSe<sub>2</sub> monolayer, which suggests a type-II band alignment. These results verify that the electronic structure of the silicane/SnSe<sub>2</sub> vdWhs is sensitive to the applied electric field.

Strain is also known as an effective regulator of the electronic structure of the vdW heterostructure [52–54]. Hence, it is now appropriate to investigate the application of an in-plane biaxial strain for the tuning of the electronic



FIGURE 4: Graphs showing (a) the band edge and (b) the bandgap as a function of the strain strength that is applied to the silicane/SnSe<sub>2</sub> vdW heterostructure.

structure of the silicane/SnSe<sub>2</sub> vdW heterostructure. Such a strain is simulated by adjusting the crystal lattice parameters. They are calculated by use of the following expression:

$$\Theta = \left[\frac{(a-a_0)}{a_0}\right] \times 100,\tag{2}$$

where  $a_0$  and a denote the unstrained and strained lattice parameters, respectively. The effects of the strain on the band edges, bandgap, and band structure of the heterostructure are shown in Figure 4 and Figure 2S. We determine that the bandgap exhibits an approximate linear increase for the strain range between -4% and 5%. Interestingly, the CBM mostly derives from the monolayer silicane, while the VBM is from the SnSe<sub>2</sub> monolayer, which signifies a type-II band alignment. Moreover, we found that the silicane/SnSe<sub>2</sub> vdW heterostructure appears to have a semiconductor-metal phase transition for a -5% strain. Therefore, effective regulation of the electronic structure of the silicane/SnSe<sub>2</sub> vdW heterostructure is plausible by applying a strain.

#### 4. Conclusion

Using first principle calculations, the electronic structure of the silicane/SnSe<sub>2</sub> vdW heterostructure is investigated when an electric field or an in-plane biaxial strain is applied. The silicane/SnSe<sub>2</sub> vdW heterostructure is demonstrated to be an indirect semiconductor for cases when an electric field is implemented within the range -1 to 0.1 V/Å, while it is metallic for 0.1 to 1 V/Å. Interestingly, at an applied electric field of -0.7 V/Å, the band alignment of the silicane/SnSe<sub>2</sub> vdW heterostructure is modified from a type-II to a type-I. Furthermore, we determine that the silicane/SnSe<sub>2</sub> vdW heterostructure has a semiconductor-metal phase transition for an applied strain of -5%. The tunability of the silicane/SnSe<sub>2</sub> vdW heterostructure bandgap enables the possibility for practical optoelectronic device applications.

#### **Data Availability**

All data, models, and code generated or used during the study appear in the submitted article.

#### **Conflicts of Interest**

There are no conflicts of interest to declare.

#### **Supplementary Materials**

*Figure 1S.* Band structures of the silicane/SnSe<sub>2</sub> vdW heterostructure when under an applied electric field with steps of 0.1 V/Å. The upper graphs from left to right range from -1.0 to -0.1 V/Å, while the lower graphs from left to right are 0.1 to 1.0 V/Å. Here, the dull-red lines represent SnSe<sub>2</sub>, while the dark blue lines denote silicane. *Figure 2S.* Band structures of the silicane/SnSe<sub>2</sub> vdW heterostructure when under inplane biaxial strains with steps of 1%. From left to right, the strains range from -5% to 5%. Here, the dull-red lines represent SnSe<sub>2</sub>, while the dark blue lines denote silicane. *(Supplementary Materials)* 

#### References

- K. S. Novoselov, A. K. Geim, S. V. Morozov et al., "Electric field effect in atomically thin carbon films," *Science*, vol. 306, p. 666, 2004.
- [2] B. Feng, Z. Ding, S. Meng et al., "Evidence of silicene in honeycomb structures of silicon on Ag(111)," *Nano Letters*, vol. 12, no. 7, pp. 3507–3511, 2012.
- [3] L. Tao, E. Cinquanta, D. Chiappe et al., "Silicene field-effect transistors operating at room temperature," *Nature Nanotechnology*, vol. 10, no. 3, pp. 227–231, 2015.
- [4] Z. Ni, Q. Liu, K. Tang et al., "Tunable bandgap in silicene and germanene," *Nano Letters*, vol. 12, no. 1, pp. 113–118, 2012.
- [5] J. Zhao, H. Liu, Z. Yu et al., "Rise of silicene: a competitive 2D material," *ProMaterials Scienceence*, vol. 83, p. 24, 2016.
- [6] H. Nakano, M. Nakano, K. Nakanishi, D. Tanaka, and T. Ohta, "Preparation of alkyl-modified silicon nanosheets by hydrosilylation of layered polysilane (Si6H6)," *Journal of the American Chemical Society*, vol. 134, p. 5452, 2012.
- [7] H. Okamoto, Y. Kumai, Y. Sugiyama et al., "Reversible C–C coupling in a uranium biheterocyclic complex," *Journal of the American Chemical Society*, vol. 132, p. 2710, 2010.
- [8] N. Gao, W. T. Zheng, and Q. Jiang, "Density functional theory calculations for two-dimensional silicene with halogen

functionalization," *Physical Chemistry Chemical Physics*, vol. 14, p. 257, 2012.

- [9] C.-W. Zhang and S.-S. Yan, "First-principles study of ferromagnetism in two-dimensional silicene with hydrogenation," *Journal of Physical Chemistry C*, vol. 116, p. 4163, 2012.
- [10] L. Li, Y. Yu, G. J. Ye et al., "Black phosphorus field-effect transistors," *Nature Nanotechnologyolocy*, vol. 9, p. 372, 2014.
- [11] H. Liu, A. T. Neal, Z. Zhu et al., "Phosphorene: an unexplored 2D semiconductor with a high hole mobility," ACS Nano, vol. 8, p. 4033, 2014.
- [12] L. Wang, X. Xu, L. Zhang et al., "Epitaxial growth of a 100square-centimetre single-crystal hexagonal boron nitride monolayer on copper," *Nature*, vol. 570, p. 91, 2019.
- [13] X. Xu, W. Yao, D. Xiao, and T. F. Heinz, "Spin and pseudospins in layered transition metal dichalcogenides," *Nature Physics*, vol. 10, p. 343, 2014.
- [14] S. Manzeli, D. Ovchinnikov, D. Pasquier, O. V. Yazyev, and A. Kis, "2D transition metal dichalcogenides," *Nature Reviews Materials*, vol. 2, p. 17033, 2017.
- [15] M. Chhowalla, Z. Liu, and H. Zhang, "Two-dimensional transition metal dichalcogenide (TMD) nanosheets," *Chemical Society Reviews*, vol. 44, p. 2584, 2015.
- [16] Q. H. Wang, K. Kalantar-Zadeh, A. Kis, J. N. Coleman, and M. S. Strano, "Electronics and optoelectronics of two-dimensional transition metal dichalcogenides," *Nature Nanotechnology*, vol. 7, p. 699, 2012.
- [17] X. Duan, C. Wang, A. Pan, R. Yu, and X. Duan, "Two-dimensional transition metal dichalcogenides as atomically thin semiconductors: opportunities and challenges," *Chemical Society Reviews*, vol. 44, p. 8859, 2015.
- [18] X. Zhou, L. Gan, W. Tian et al., "Ultrathin SnSe2 flakes grown by chemical vapor deposition for high-performance photodetectors," *Advanced Materials*, vol. 27, p. 8035, 2015.
- [19] F. Zhang, C. Xia, J. Zhu et al., "SnSe2 2D anodes for advanced sodium ion batteries," *Advanced Energy Materials*, vol. 6, Article ID 1601188, 2016.
- [20] T. Roy, M. Tosun, M. Hettick, G. H. Ahn, C. Hu, and A. Javey, "2D-2D tunneling field-effect transistors using WSe<sub>2</sub>/SnSe<sub>2</sub> heterostructures," *Applied Physics Letters*, vol. 108, Article ID 083111, 2016.
- [21] J. M. Gonzalez and I. I. Oleynik, "Layer-dependent properties of SnS2 and SnSe<sub>2</sub> novel two-dimensional materials," *Physical Review B: Condensed Matter*, vol. 94, Article ID 125443, 2016.
- [22] X. Zhou, N. Zhou, C. Li et al., "Vertical heterostructures based on SnSe<sub>2</sub>/MoS2 for high performance photodetectors," 2D Materials, vol. 4, Article ID 25048, 2017.
- [23] L. Liu, Y. Feng, and Z. Shen, "Structural and electronic properties of h-BN," *Physical Review B: Condensed Matter*, vol. 68, Article ID 104102, 2003.
- [24] F. Schedin, A. K. Geim, S. V. Morozov et al., "Detection of individual gas molecules adsorbed on graphene," *Nature Materials*, vol. 6, p. 652, 2007.
- [25] A. Preobrajenski, M. Nesterov, M. L. Ng, A. Vinogradov, and N. Mårtensson, "Monolayer h-BN on lattice-mismatched metal surfaces: on the formation of the nanomesh," *Chemical Physics Letters*, vol. 446, p. 119, 2007.
- [26] A. K. Geim and I. V. Grigorieva, "Van der Waals heterostructures," *Nature*, vol. 499, p. 419, 2013.
- [27] Y. Liu, N. O. Weiss, X. Duan, H.-C. Cheng, Y. Huang, and X. Duan, "Van der Waals heterostructures and devices," *Nature Reviews Materials*, vol. 1, p. 1, 2016.
- [28] K. Novoselov, A. Mishchenko, A. Carvalho, and A. C. Neto, "2D materials and van der Waals heterostructures," *Science*, vol. 353, p. 9493, 2016.

- [29] D. Jariwala, T. J. Marks, and M. C. Hersam, "Mixed-dimensional van der Waals heterostructures," *Nature Materials*, vol. 16, p. 170, 2017.
- [30] R.-w. Zhang, C.-w. Zhang, W.-x. Ji et al., "Silicane as an inert substrate of silicene: a promising candidate for FET," *Journal* of *Physical Chemistry C*, vol. 118, p. 25278, 2014.
- [31] S. Yamanaka, H. Matsu-ura, and M. Ishikawa, "New deintercalation reaction of calcium from calcium disilicide. synthesis of layered polysilane," *Materials Research Bulletin*, vol. 31, p. 307, 1996.
- [32] O. D. Restrepo, R. Mishra, J. E. Goldberger, and W. Windl, "Tunable gaps and enhanced mobilities in strain-engineered silicane," *Journal of Applied Physics*, vol. 115, Article ID 33711, 2014.
- [33] C. Gang, L. Peng-Fei, and L. Zi-Tao, "Optimization of a magneto-optic trap using nanofibers," *Chinese Physics B*, vol. 22, Article ID 46201, 2013.
- [34] S. Wang and J. Yu, "Tuning electronic properties of silicane layers by tensile strain and external electric field: A firstprinciples study," *Thin Solid Films*, vol. 654, p. 107, 2018.
- [35] Y. Huang, C. Ling, H. Liu, S. Wang, and B. Geng, "Versatile electronic and magnetic properties of SnSe2 nanostructures induced by the strain," *Journal of Physical Chemistry C*, vol. 118, p. 9251, 2014.
- [36] N. D. Hien, N. Q. Cuong, L. M. Bui et al., "Physica E: lowdimensional systems and nanostructures," *Physica A*, vol. 111, p. 201, 2019.
- [37] C. Xia, J. Du, M. Li et al., "Effects of electric field on the electronic structures of broken-gap phosphorene/Sn X2 (X = S, Se) van der waals heterojunctions," *Physicals Reviews Applied*, vol. 10, Article ID 54064, 2018.
- [38] J. P. Perdew and Y. Wang, "Accurate and simple analytic representation of the electron-gas correlation energy," *Physical Review B: Condensed Matter*, vol. 45, Article ID 13244, 1992.
- [39] J. P. Perdew, K. Burke, and M. Ernzerhof, "Generalized gradient approximation made simple," *Physical Review Letters*, vol. 77, p. 3865, 1996.
- [40] M. Brandbyge, J.-L. Mozos, P. Ordejón, J. Taylor, and K. Stokbro, "Density-functional method for nonequilibrium electron transport," *Physical Review B: Condensed Matter*, vol. 65, Article ID 165401, 2002.
- [41] J. M. Soler, E. Artacho, and J. D Gale, "The SIESTA method; developments and applicability," *Journal of Physics: Condensed Matter*, vol. 14, pp. 2745–2779, 2002.
- [42] D. Chakraborty, K. Berland, and T. Thonhauser, "Nextgeneration nonlocal van der waals density functional," *American Physical.Society*, vol. 16, 2020.
- [43] G. Guzmán-Verri and L. L. Y. Voon, "Band structure of hydrogenated Si nanosheets and nanotubes," *Journal of Physics: Condensed Matter*, vol. 23, Article ID 145502, 2011.
- [44] G. Li, G. Ding, and G. Gao, "Thermoelectric properties of SnSe2 monolayer," *Journal of Physics: Condensed Matter*, vol. 29, Article ID 15001, 2016.
- [45] X. Tong, L. Fang, and R. Liu, "Strain-induced electronic properties of van der Waals heterostructures based on tin dichalcogenides," *AIP Advances*, vol. 9, Article ID 55324, 2019.
- [46] Z. Guan, C.-S. Lian, S. Hu, S. Ni, J. Li, and W. Duan, "Tunable structural, electronic, and optical properties of layered twodimensional C2N and MoS2 van der waals heterostructure as photovoltaic material," *Journal of Physical Chemistry C*, vol. 121, Article ID 36654, 2017.

- [47] K. D. Pham, N. N. Hieu, L. M. Bui et al., "Vertical strain and electric field tunable electronic properties of type-II band," *Chemical Physics Letters*, vol. 716, p. 155, 2019.
- [48] C. Li, X. Yan, X. Song et al., "WSe 2/MoS 2 and MoTe 2/SnSe 2 van der Waals heterostructure transistors with different band alignment," *Nanotechnology*, vol. 28, Article ID 415201, 2017.
- [49] V. OngunÖzçelik, J. G. Azadani, C. Yang, S. J. Koester, and T. Low, "Band alignment of two-dimensional semiconductors for designing heterostructures with momentum space matching," *Physical Review B: Condensed Matter*, vol. 94, Article ID 35125, 2016.
- [50] Z. Ma, Y. Wang, Y. Wei, C. Li, X. Zhang, and F. Wang, "A type-II C2N/α-Te van der Waals heterojunction with improved optical properties by external perturbation," *Physical Chemistry Chemical Physics*, vol. 21, Article ID 21753, 2019.
- [51] H. T. Nguyen, T. V. Vu, N. N. Hieu et al., "Effects of electric field and strain engineering on the electronic properties, band alignment and enhanced optical properties of ZnO/Janus ZrSSe heterostructures," *RSC Advances*, vol. 10, p. 2967, 2020.
- [52] N. Song, H. Ling, Y. Wang, L. Zhang, Y. Yang, and Y. Jia, "Intriguing electronic properties of germanene/indium selenide and antimonene/indium selenide heterostructures," *Journal of Solid State Chemistry*, vol. 269, p. 513, 2019.
- [53] X.-H. Li, B.-J. Wang, X.-L. Cai et al., "Strain-tunable electronic properties and band alignments in GaTe/C2N heterostructure: a first-principles calculation," *Nano Research Letter*, vol. 13, p. 300, 2018.
- [54] X. Wang, Y. Wang, R. Quhe, Y. Tang, X. Dai, and W. Tang, "Designing strained C2N/GaTe(InTe) heterostructures for photovoltaic and photocatalytic application," *Journal of Alloy* and Compounds, vol. 816, Article ID 152559, 2020.